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RESEARCH MEMORANDUM

ELEVATED-TEMPERATURE FATIGUE PROPERTIES OF
TWO TITANIUM ALLOYS

By William K. Rey

University of Alabama

NATIONAL ADVISORY COMMITTEE
FOR AERONAUTICS

WASHINGTON

April 24, 1956

NACA RM 56B07

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RESEARCH MEMORANDUM

ELEVATED-TEMPERATURE FATIGUE PROPERTIES OF
TWO TITANIUM ALLOYS

By William K. Rey

SUMMARY

An investigation was conducted to evaluate the unnotched fatigue properties of 3Mn Complex and 3Al-5Cr titanium alloys at elevated temperatures. Fatigue studies were conducted for each alloy at room temperature, 200°, 400°, 600°, 800°, and 1,000° F. The results are presented in tabular form and as curves of stress versus cycles to failure for each test temperature. The endurance strength at 10,000,000 cycles for the 3Mn Complex alloy decreased from 79,000 psi at room temperature to 26,500 psi at 1,000° F. The endurance strength at 10,000,000 cycles for the 3Al-5Cr alloy decreased from 91,000 psi at room temperature to 46,500 psi at 1,000° F. The decrease in endurance strength with an increase in temperature is shown by a curve of endurance strength versus temperature for each alloy.

INTRODUCTION

During the past few years metallurgical research has provided the engineer with alloys of titanium that are taking their place as important structural materials. These alloys are of particular interest to the aircraft industry since they possess a unique combination of mechanical properties - lightness, high strength, general resistance to environmental attack, and retention of strength at moderately elevated temperatures. To make the most effective use of these alloys, it will be necessary for the designer to have available the mechanical properties for various types of loading under different environmental conditions.

For many applications, the behavior of a material when it is subjected to repeated stressing is of prime importance. This is true since many of the structural components are subjected to repeated loading and unloading. This investigation was undertaken to determine the unnotched fatigue properties of two titanium alloys at temperatures up to 1,000° F because of the potential use of titanium alloys in this temperature range.

This investigation was initiated under the sponsorship of the University Research Committee of the University of Alabama and completed with the University Research Committee and the National Advisory Committee for Aeronautics as cosponsors. The University Research Committee supplied funds for the necessary equipment and the National Advisory Committee for Aeronautics furnished the operating funds. The material required for preparation of the test specimens was donated by the Mallory-Sharon Titanium Corporation of Niles, Ohio.

MATERIAL

The alloy designated 3Mn Complex titanium alloy was supplied as hot-rolled and cleaned 1/2-inch-diameter round rod with all material coming from the same heat. The chemical composition by weight of this heat as determined by the Mallory-Sharon laboratory was as follows:

| | |
|-------------------------------|-------|
| Carbon, percent | 0.03 |
| Nitrogen, percent | 0.010 |
| Hydrogen, percent | 0.012 |
| Iron, percent | 0.93 |
| Manganese, percent | 3.34 |
| Chromium, percent | 1.07 |
| Vanadium, percent | 1.03 |
| Molybdenum, percent | 1.01 |
| Titanium | Bal. |

The room-temperature mechanical properties were determined using American Society for Metals standard 5/16-inch tension specimens. These tests were performed in a Baldwin 60,000-pound universal testing machine with a Huggenberger Tensometer used to measure strains. The average room-temperature mechanical properties from three tests were as follows:

| | |
|--|------------|
| Ultimate strength, psi | 147,900 |
| Proportional limit, psi | 126,000 |
| Yield strength (0.2-percent offset), psi | 134,750 |
| Young's modulus, psi | 16,200,000 |
| Elongation in 1 inch, percent | 24 |
| Reduction of area, percent | 57.8 |
| Rockwell hardness | 33.8C |

The average tensile stress-strain curve for the 3Mn Complex alloy is shown in figure 1.

The second alloy, which is designated 3Al-5Cr titanium alloy, was also supplied as hot-rolled and cleaned 1/2-inch-diameter round rod with all material coming from the same heat. The chemical composition by weight as determined by the Mallory-Sharon laboratory was as follows:

| | |
|-----------------------------|-------|
| Carbon, percent | 0.05 |
| Nitrogen, percent | 0.036 |
| Hydrogen, percent | 0.011 |
| Iron, percent | 0.25 |
| Aluminum, percent | 3.47 |
| Chromium, percent | 4.94 |
| Titanium | Bal. |

The room-temperature mechanical properties were determined by the same procedure used for the other alloy. The average room-temperature mechanical properties for the 3Al-5Cr alloy were as follows:

| | |
|--|------------|
| Ultimate strength, psi | 140,800 |
| Proportional limit, psi | 111,800 |
| Yield strength (0.2-percent offset), psi | 125,300 |
| Young's modulus, psi | 15,500,000 |
| Elongation in 1 inch, percent | 21 |
| Reduction of area, percent | 57.8 |
| Rockwell hardness | 35.5C |

The average tensile stress-strain curve for the 3Al-5Cr alloy is shown in figure 2.

APPARATUS AND PROCEDURE

Figure 3 shows the Krouse high-speed, high-temperature, repeated-stress machine used for all fatigue tests. This machine loads the specimen as a simple beam with a constant bending moment throughout the length of the specimen. It is equipped with a Marshall furnace and Foxboro potentiometer controller that permit testing at room temperature and in the range from 200° to 1,800° F with an accuracy of $\pm 2^\circ$ F.

Prior to testing, it was necessary to perform a load calibration to determine the load necessary to balance the weight of the driving motors and specimen holders. This was accomplished by using a dummy specimen to which two type A-8 SR-4 electric strain gages were attached. Loads were applied in 1-pound increments and the strains determined for each load. Curves of load versus strain were plotted to determine the tare load. This load calibration was confirmed by testing a number of stainless-steel specimens in this machine and comparing the results with data obtained from another machine. This check showed excellent agreement in the results obtained from the two machines.

The furnace temperature is controlled during testing by means of a Chromel-Alumel thermocouple placed at the center of the furnace midway between the specimen and the furnace wall. To determine the correlation between the temperature of this control thermocouple and the specimen temperature, an iron-constantan thermocouple was attached to the center of a specimen. For each of the test temperatures, a series of readings was taken to determine the difference in temperature at the two thermocouple locations. These data showed that after temperature equilibrium was reached there was a maximum of 2° F difference in temperature at the two locations. An additional investigation showed that the temperature was constant throughout the length of the specimen.

The dimensions of the specimens used for all fatigue tests are given in figure 4. These specimens were prepared from 1/2-inch-diameter rod and then polished. The machining marks were removed with 3/0 emery cloth, and 400-A Durite paper was used for the final polish. All circumferential scratches were removed by polishing parallel to the longitudinal axis of the specimen while it slowly rotated in a lathe. Approximately 0.002 inch of the material was removed during the polishing operation.

The specimens were inserted in the furnace at room temperature and rotated at zero stress while the furnace temperature was increased to the test temperature. The testing temperature was attained in 45 minutes. An additional 15 minutes was allowed to obtain temperature equilibrium before applying the load. All tests were conducted at a speed of 4,800 cycles per minute. The test temperatures were room temperature, 200°, 400°, 600°, 800°, and 1,000° F.

RESULTS AND DISCUSSION

The results of the fatigue tests of the 3Mn Complex alloy are presented in tabular form in table I and as curves of nominal stress versus cycles to failure in figures 5(a) to 5(f). The endurance strengths at 10,000,000 cycles from these curves, are compared in the following table:

| Temperature, °F | Endurance strength, psi | Endurance ratio |
|------------------|-------------------------|-----------------|
| Room temperature | 79,000 | 0.53 |
| 200 | 65,500 | .44 |
| 400 | 64,000 | .43 |
| 600 | 55,500 | .38 |
| 800 | 45,000 | .30 |
| 1,000 | 26,500 | .18 |

The endurance ratios in this table were computed as the ratio of the endurance strength at 10,000,000 cycles to the ultimate strength at room temperature. Although this ratio is not the true endurance ratio for the elevated temperatures, it is a measure of the reduction in strength as temperature increases. As shown in the table, the endurance strength decreased from 79,000 psi at room temperature to 26,500 psi at 1,000° F. The reduction in strength is shown graphically by a curve of endurance strength versus temperature in figure 6.

The curves of stress versus cycles to failure for the 3Mn Complex alloy exhibit small scatter at room temperature, 200°, and 800° F. While the data at 400° and 600° F show greater scatter, it is not unreasonable. The small number of specimens available for testing at 1,000° F was due to the limited amount of available material. However, a sufficient number of tests were performed at 1,000° F to give a reasonable indication of the endurance strength at this temperature.

The results of the fatigue tests of the 3Al-5Cr alloy are presented in tabular form in table II and as curves of nominal stress versus cycles to failure in figures 7(a) to 7(f). The endurance strengths at 10,000,000 cycles from these curves are compared in the following table:

| Temperature, °F | Endurance strength, psi | Endurance ratio |
|------------------|-------------------------|-----------------|
| Room temperature | 91,000 | 0.65 |
| 200 | 84,000 | .60 |
| 400 | 79,500 | .56 |
| 600 | 73,400 | .52 |
| 800 | 62,250 | .44 |
| 1,000 | 46,500 | .33 |

The endurance ratio was computed as for the 3Mn Complex alloy. The endurance strength decreased from 91,000 psi at room temperature to 46,500 psi at 1,000° F. The curve of endurance strength versus temperature is shown in figure 6.

Some of the scatter in the fatigue results may be attributed to the fact that neither material was annealed after rolling. Since the temperature calibration was performed under static conditions, it is possible that the rotation of the specimen produced a small temperature change that would further account for the scatter in the test results.

In table III the ratios of endurance strength to weight of the two titanium alloys and four aluminum alloys are compared at four temperatures. The endurance strengths of the titanium alloys at 300° and 500° F were obtained from figure 6 by interpolation. The endurance strengths of the

aluminum alloys were obtained from reference 1. This comparison shows that the titanium alloys are superior to the aluminum alloys at all four temperatures on the basis of their ratios of endurance strength to weight. The 3Al-5Cr alloy has a higher ratio of endurance strength to weight at all temperatures than the 3Mn Complex alloy even though it has a lower ultimate tensile strength at room temperature.

It is of interest to note that the curves of endurance strength versus temperature have the same shape for both materials. The small reduction in endurance strength for the 3Mn Complex alloy between 200° and 600° F is surprising when compared with the reduction in endurance strength between room temperature and 200° F. In plotting these curves, the room temperature was taken as 75° F.

CONCLUDING REMARKS

Within the limitations of test scatter, the results of a study of the fatigue properties of two titanium alloys show that both of the alloys have potential use in the temperature range investigated. The 3Al-5Cr alloy has a higher endurance strength than the 3Mn Complex alloy at all temperatures considered in this study even though it has a lower ultimate tensile strength at room temperature.

A comparison of the two titanium alloys with aluminum alloys shows that the titanium alloys are superior on the basis of their ratios of endurance strength to weight.

Further study is needed to complete the evaluation of these alloys. A study of the possible correlation between the endurance strength at elevated temperatures and the stress to rupture at these temperatures would be of value. An investigation of the notch sensitivity at elevated temperatures is also necessary to complete the evaluation for applications involving repeated stressing.

University of Alabama,
University, Ala., May 12, 1955.

REFERENCE

1. Anon.: Strength of Metal Aircraft Elements. ANC-5, Munitions Board Aircraft Committee, Mar. 1955.

TABLE I.- RESULTS OF FATIGUE TESTS OF 3Mn COMPLEX TITANIUM ALLOY

| Specimen | Stress, psi | Cycles to failure | Remarks |
|---------------------|-------------|-------------------|--------------|
| At room temperature | | | |
| 10F 2 | 100,720 | 7,300 | |
| 10F 16 | 100,280 | 10,700 | |
| 10F 15 | 98,320 | 16,800 | |
| 10F 12 | 98,030 | 18,500 | |
| 10F 3 | 94,270 | 20,600 | |
| 10F 4 | 90,370 | 28,700 | |
| 10F 11 | 87,910 | 64,800 | |
| 10F 5 | 86,320 | 97,700 | |
| 10F 6 | 84,760 | 61,100 | |
| 10F 7 | 82,030 | 144,900 | |
| 10F 8 | 79,980 | 450,200 | |
| 10F 14 | 79,640 | 21,713,800 | Did not fail |
| 10F 10 | 79,230 | 340,400 | |
| 10F 9 | 78,460 | 14,347,500 | Did not fail |
| At 200° F | | | |
| 10F 17 | 83,830 | 40,100 | |
| 10F 18 | 81,070 | 24,800 | |
| 10F 19 | 80,580 | 47,000 | |
| 10F 20 | 76,590 | 44,400 | |
| 10F 21 | 72,320 | 88,000 | |
| 10F 25 | 70,650 | 318,500 | |
| 10F 22 | 69,210 | 704,800 | |
| 10F 24 | 67,430 | 2,272,100 | |
| 10F 27 | 66,520 | 3,618,100 | |
| 10F 26 | 65,930 | 1,710,000 | |
| 10F 31 | 65,900 | 13,205,100 | Did not fail |
| 10F 29 | 65,080 | 1,711,900 | |
| 10F 28 | 64,500 | 1,456,500 | |
| 10F 30 | 64,350 | 10,120,800 | Did not fail |
| 10F 23 | 64,230 | 12,065,200 | Did not fail |
| At 400° F | | | |
| 10F 38 | 71,930 | 32,200 | |
| 10F 41 | 70,060 | 165,100 | |
| 10F 43 | 69,980 | 46,500 | |
| 10F 39 | 69,160 | 1,262,100 | |
| 10F 35 | 68,700 | 402,400 | |
| 10F 42 | 68,010 | 38,400 | |
| 10F 44 | 67,590 | 34,300 | |
| 10F 36 | 66,760 | 173,300 | |
| 10F 40 | 66,400 | 80,400 | |
| 10F 37 | 65,380 | 47,800 | |
| 10F 48 | 65,020 | 2,003,500 | |
| 10F 34 | 64,670 | 22,804,000 | Did not fail |
| 10F 49 | 64,140 | 10,651,900 | |
| 10F 46 | 64,070 | 1,028,200 | |
| 10F 47 | 63,020 | 10,286,600 | Did not fail |
| 10F 45 | 61,990 | 10,041,700 | Did not fail |

TABLE I.- RESULTS OF FATIGUE TESTS OF 3Mn COMPLEX TITANIUM ALLOY - Concluded

| Specimen | Stress, psi | Cycles to failure | Remarks |
|-------------|-------------|-------------------|--------------|
| At 600° F | | | |
| 10F 51 | 84,060 | 4,100 | |
| 10F 74 | 64,000 | 23,200 | |
| 10F 58 | 63,750 | 20,300 | |
| 10F 57 | 62,900 | 135,400 | |
| 10F 59 | 62,870 | 116,200 | |
| 10F 73 | 62,820 | 50,200 | |
| 10F 60 | 62,200 | 30,600 | |
| 10F 61 | 61,530 | 583,900 | |
| 10F 62 | 61,000 | 129,300 | |
| 10F 75 | 60,930 | 156,100 | |
| 10F 63 | 60,490 | 768,200 | |
| 10F 76 | 60,080 | 200,000 | |
| 10F 64 | 59,950 | 1,106,300 | |
| 10F 65 | 59,490 | 305,400 | |
| 10F 66 | 59,060 | 130,700 | |
| 10F 67 | 59,060 | 353,900 | |
| 10F 68 | 57,900 | 1,542,300 | |
| 10F 69 | 57,250 | 2,598,400 | |
| 10F 70 | 56,380 | 3,441,100 | |
| 10F 72 | 54,950 | 12,090,500 | Did not fail |
| 10F 52 | 43,060 | 12,406,400 | Did not fail |
| At 800° F | | | |
| 10F 81 | 59,030 | 12,200 | |
| 10F 82 | 57,870 | 16,200 | |
| 10F 83 | 57,000 | 14,800 | |
| 10F 84 | 55,960 | 28,100 | |
| 10F 85 | 55,020 | 28,200 | |
| 10F 86 | 53,880 | 34,400 | |
| 10F 87 | 53,190 | 57,500 | |
| 10F 89 | 52,040 | 61,900 | |
| 10F 90 | 51,020 | 102,500 | |
| 10F 91 | 50,000 | 186,900 | |
| 10F 93 | 49,490 | 138,300 | |
| 10F 94 | 49,080 | 209,700 | |
| 10F 92 | 48,160 | 1,905,300 | |
| 10F 97 | 47,990 | 664,300 | |
| 10F 95 | 47,540 | 652,300 | |
| 10F 98 | 47,060 | 335,700 | |
| 10F 99 | 46,010 | 888,700 | |
| 10F 101 | 45,460 | 1,765,300 | |
| 10F 100 | 45,040 | 10,156,700 | |
| At 1,000° F | | | |
| 10F 77 | 54,790 | 7,800 | |
| 10F 78 | 34,960 | 214,600 | |
| 10F 80 | 33,500 | 286,100 | |
| 10F 79 | 32,010 | 3,131,000 | |
| 10F 102 | 31,000 | 434,900 | |
| 10F 104 | 29,000 | 1,092,300 | |
| 10F 103 | 26,370 | 12,318,800 | Did not fail |

TABLE II.- RESULTS OF FATIGUE TESTS OF 3A-5C TITANIUM ALLOY

| Specimen | Stress, psi | Cycles to failure | Remarks |
|---------------------|-------------|-------------------|--------------|
| At room temperature | | | |
| 9F 6 | 99,630 | 26,800 | |
| 9F 7 | 98,240 | 19,600 | |
| 9F 12 | 97,000 | 23,200 | |
| 9F 4 | 95,280 | 36,400 | |
| 9F 13 | 95,020 | 29,000 | |
| 9F 40 | 94,460 | 57,100 | |
| 9F 41 | 94,300 | 105,700 | |
| 9F 38 | 94,090 | 1,017,000 | |
| 9F 39 | 94,090 | 12,786,500 | Did not fail |
| 9F 14 | 92,870 | 57,400 | |
| 9F 2 | 92,490 | 10,083,200 | Did not fail |
| 9F 18 | 91,720 | 66,300 | |
| 9F 19 | 91,480 | 65,400 | |
| 9F 17 | 91,450 | 54,600 | |
| 9F 20 | 91,060 | 54,757,300 | Did not fail |
| 9F 16 | 90,710 | 13,466,000 | Did not fail |
| 9F 15 | 87,120 | 13,000,000 | Did not fail |
| 9F 1 | 81,220 | 21,322,500 | Did not fail |
| At 200° F | | | |
| 9F 21 | 92,860 | 58,300 | |
| 9F 22 | 90,150 | 72,100 | |
| 9F 25 | 89,290 | 84,300 | |
| 9F 23 | 87,650 | 45,900 | |
| 9F 26 | 87,220 | 112,700 | |
| 9F 35 | 86,000 | 57,000 | |
| 9F 33 | 85,860 | 75,600 | |
| 9F 30 | 85,770 | 65,700 | |
| 9F 29 | 85,770 | 77,100 | |
| 9F 36 | 85,490 | 184,200 | |
| 9F 37 | 85,290 | 59,500 | |
| 9F 34 | 85,000 | 14,160,500 | Did not fail |
| 9F 28 | 84,860 | 71,500 | |
| 9F 32 | 84,760 | 101,549,500 | Did not fail |
| 9F 31 | 84,490 | 51,800 | |
| 9F 27 | 84,010 | 104,550,400 | Did not fail |
| At 400° F | | | |
| 9F 56 | 89,800 | 28,200 | |
| 9F 57 | 87,860 | 36,800 | |
| 9F 53 | 84,810 | 93,500 | |
| 9F 54 | 84,650 | 45,200 | |
| 9F 58 | 83,870 | 27,200 | |
| 9F 42 | 82,080 | 72,400 | |
| 9F 60 | 81,110 | 65,200 | |
| 9F 59 | 80,870 | 25,200 | |
| 9F 50 | 80,250 | 99,972,400 | Did not fail |
| 9F 49 | 79,900 | 2,284,400 | |
| 9F 52 | 79,230 | 2,092,900 | |
| 9F 48 | 75,550 | 22,288,900 | Did not fail |
| 9F 47 | 75,030 | 12,757,500 | Did not fail |
| 9F 46 | 74,330 | 14,190,700 | Did not fail |
| 9F 45 | 73,240 | 19,498,300 | Did not fail |

TABLE II.- RESULTS OF FATIGUE TESTS OF 3A-5C TITANIUM ALLOY - Concluded

| Specimen | Stress, psi . | Cycles to failure | Remarks |
|-------------|---------------|-------------------|--------------|
| At 600° F | | | |
| 9F 74 | 76,110 | 8,183,100 | |
| 9F 75 | 76,070 | 149,300 | |
| 9F 62 | 75,960 | 37,800 | |
| 9F 73 | 75,200 | 76,400 | |
| 9F 63 | 74,930 | 40,000 | |
| 9F 71 | 74,600 | 98,300 | |
| 9F 72 | 74,240 | 4,587,000 | |
| 9F 61 | 74,140 | 4,325,300 | |
| 9F 64 | 74,090 | 34,900 | |
| 9F 70 | 73,870 | 14,190,700 | Did not fail |
| 9F 69 | 73,690 | 13,093,100 | Did not fail |
| 9F 68 | 73,520 | 4,663,400 | |
| 9F 66 | 72,980 | 10,216,800 | Did not fail |
| 9F 65 | 71,750 | 13,975,000 | Did not fail |
| At 800° F | | | |
| 9F 80 | 74,000 | 40,300 | |
| 9F 77 | 72,990 | 22,900 | |
| 9F 82 | 71,990 | 27,700 | |
| 9F 90 | 71,000 | 25,600 | |
| 9F 76 | 70,040 | 76,600 | |
| 9F 89 | 69,990 | 19,100 | |
| 9F 81 | 68,990 | 22,800 | |
| 9F 85 | 67,960 | 57,100 | |
| 9F 84 | 67,000 | 26,600 | |
| 9F 94 | 66,510 | 126,000 | |
| 9F 83 | 65,990 | 82,600 | |
| 9F 91 | 64,990 | 148,400 | |
| 9F 78 | 64,990 | 187,200 | |
| 9F 86 | 64,010 | 78,500 | |
| 9F 92 | 63,500 | 90,400 | |
| 9F 88 | 63,010 | 271,200 | |
| 9F 93 | 62,510 | 11,859,200 | Did not fail |
| 9F 87 | 61,990 | 4,996,500 | |
| At 1,000° F | | | |
| 9F 96 | 56,990 | 42,000 | |
| 9F 95 | 54,990 | 250,500 | |
| 9F 97 | 53,000 | 417,000 | |
| 9F 98 | 50,510 | 481,900 | |
| 9F 100 | 50,000 | 2,406,100 | |
| 9F 101 | 49,000 | 166,200 | |
| 9F 102 | 47,490 | 425,000 | |
| 9F 103 | 46,570 | 12,222,900 | Did not fail |
| 9F 99 | 45,000 | 11,088,900 | Did not fail |

TABLE III.- COMPARISON OF RATIOS OF ENDURANCE STRENGTH^a TO WEIGHT

| Material | Weight, W, lb/cu in. | At room temp. | | At 300° F | | At 400° F | | At 500° F | |
|------------------------|-------------------------|----------------|-------------------|----------------|-------------------|----------------|-------------------|----------------|-------------------|
| | | F _e | F _e /W | F _e | F _e /W | F _e | F _e /W | F _e | F _e /W |
| 3Mn Ti alloy | 0.170 | 79,000 | 464,700 | 65,000 | 382,400 | 64,000 | 376,500 | 61,500 | 361,800 |
| 3Al-5Cr Ti alloy | .166 | 91,000 | 548,000 | 80,500 | 484,900 | 79,500 | 478,900 | 77,000 | 463,900 |
| 2014-T6 aluminum alloy | .101 | 24,000 | 237,600 | 15,000 | 148,500 | 10,000 | 99,000 | 7,000 | 69,300 |
| 2024-T4 aluminum alloy | .100 | 24,000 | 240,000 | 17,000 | 170,000 | 13,000 | 130,000 | 8,500 | 85,000 |
| 6061-T6 aluminum alloy | .098 | 17,000 | 173,500 | 14,000 | 142,900 | 11,000 | 112,200 | 5,500 | 56,100 |
| 7075-T6 aluminum alloy | .101 | 24,000 | 237,600 | 13,000 | 128,700 | 9,500 | 94,100 | 8,000 | 79,200 |

^aIn this table, the endurance strength F_e is taken at 10,000,000 cycles.

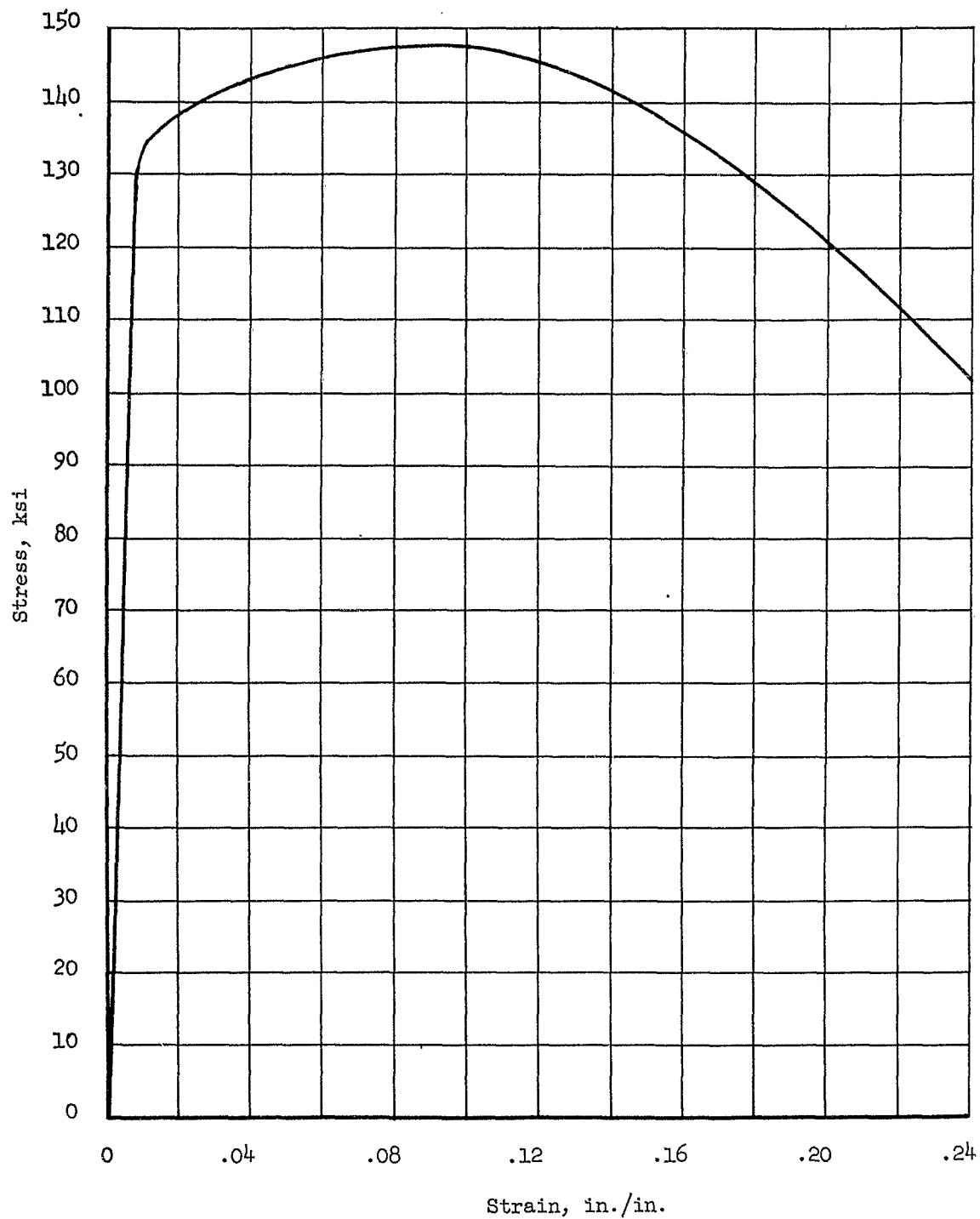


Figure 1.- Tensile stress-strain curve for 3Mn Complex titanium alloy at room temperature.

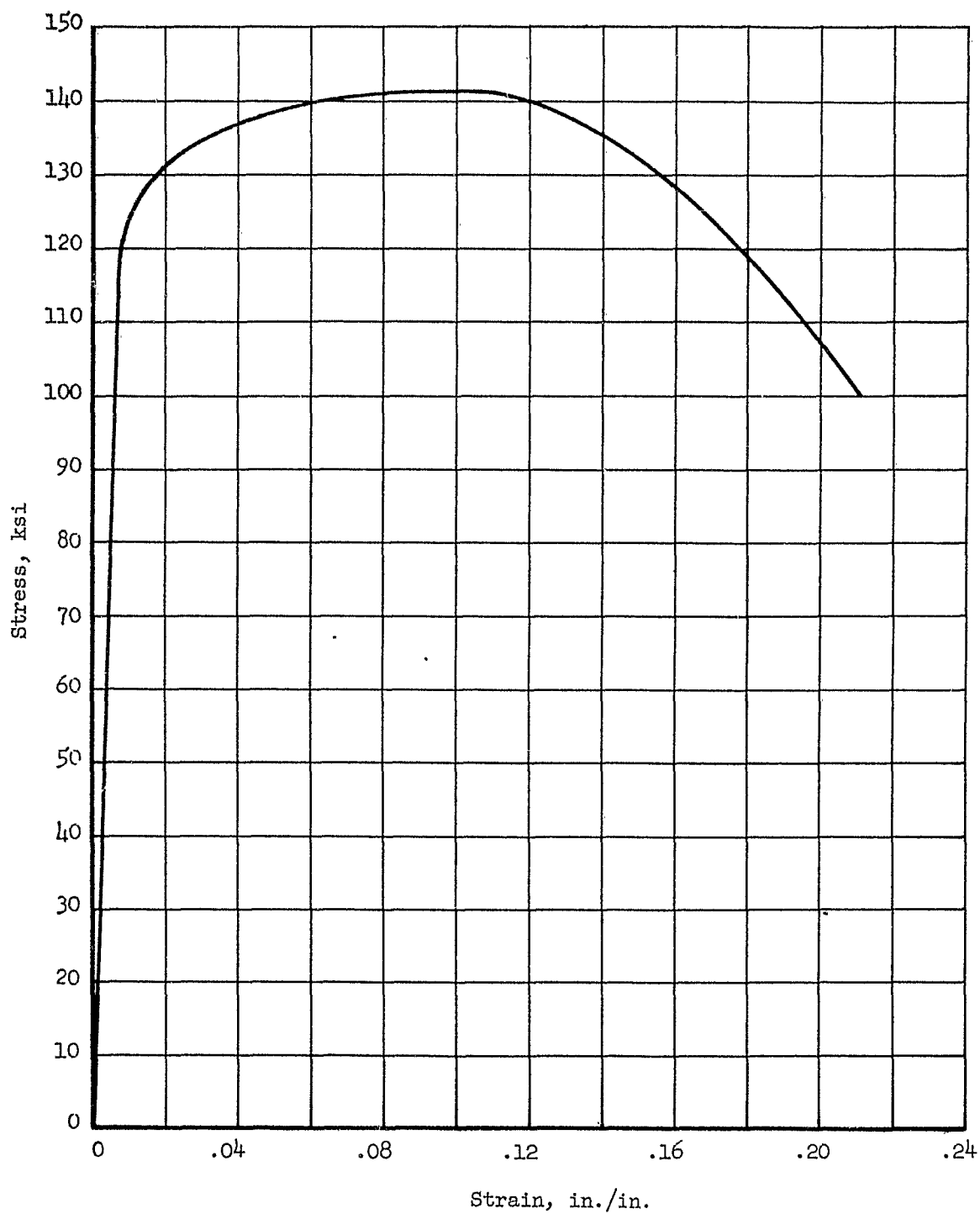
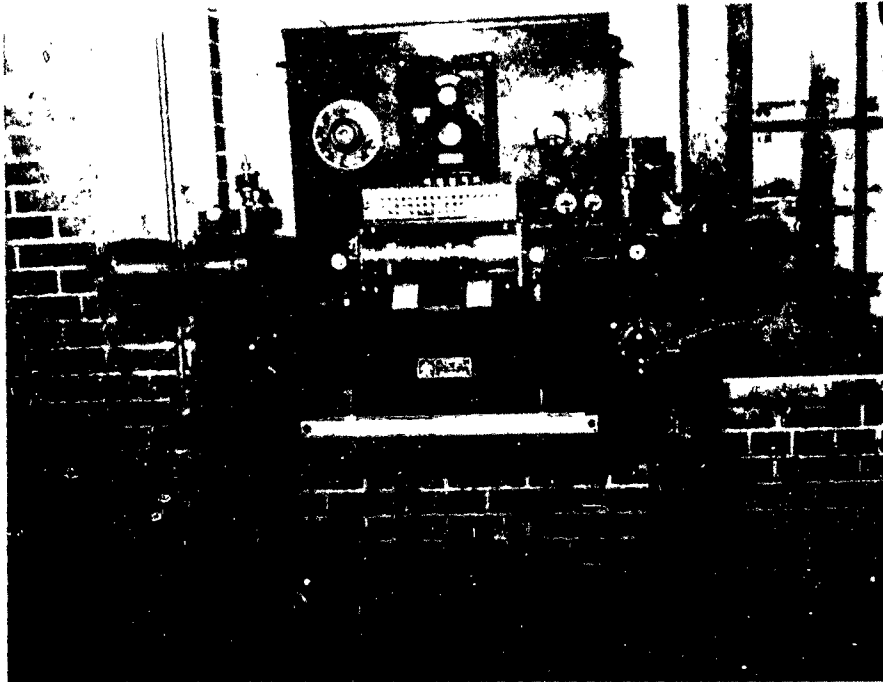


Figure 2.- Tensile stress-strain curve for 3Al-5Cr titanium alloy at room temperature.



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Figure 3.- Krouse high-speed, high-temperature, rotating-beam fatigue machine.

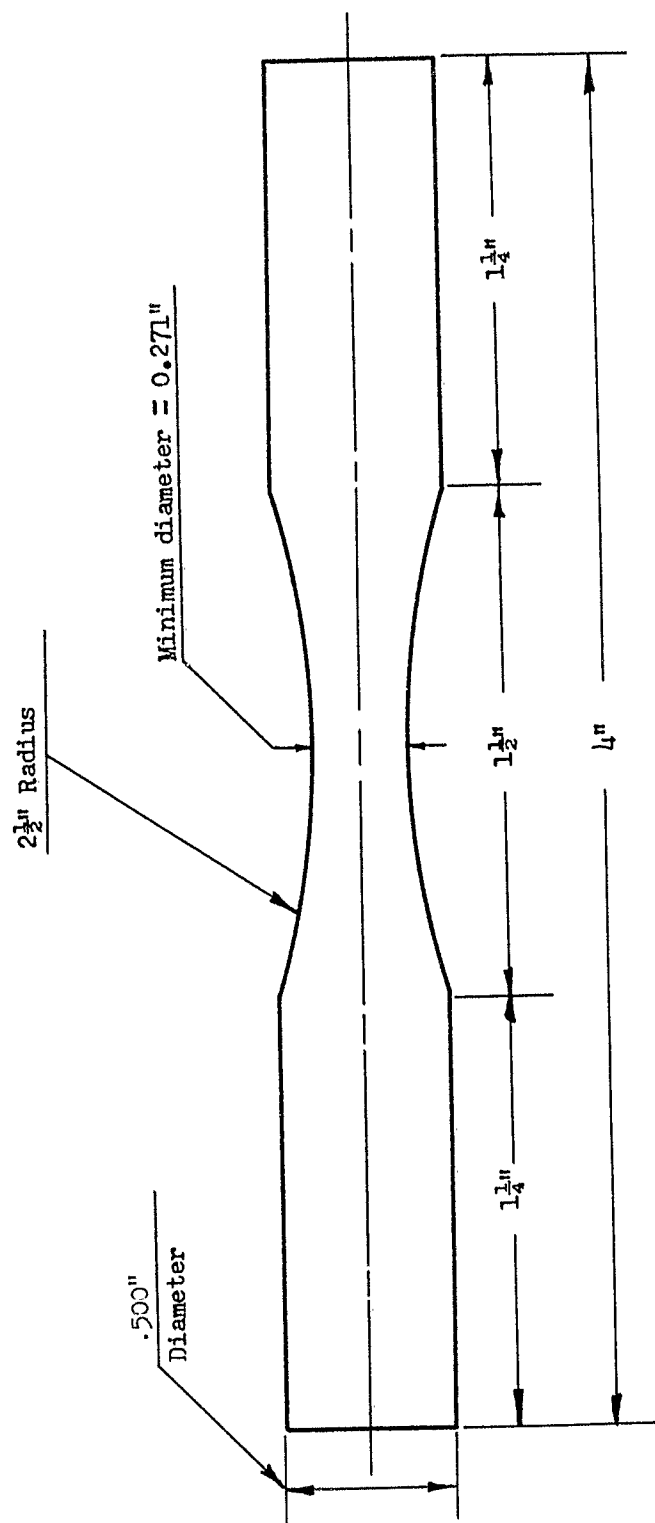
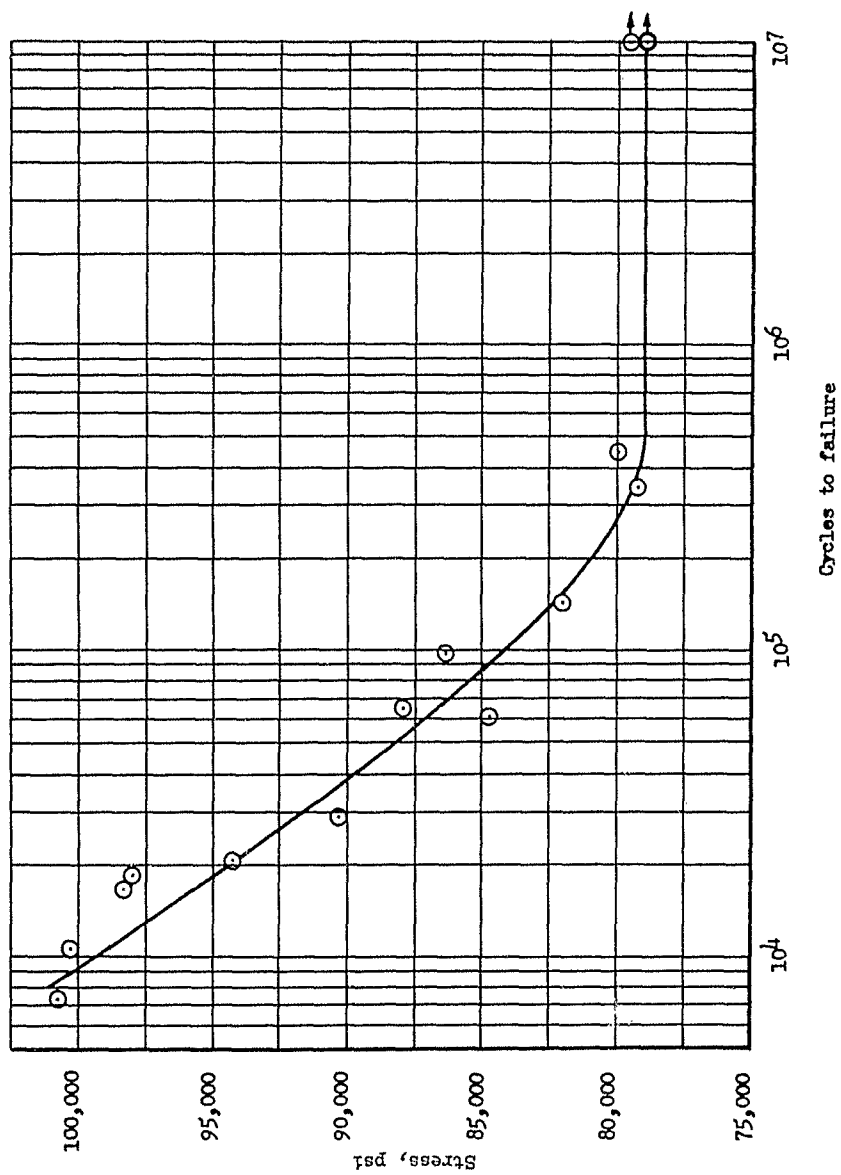
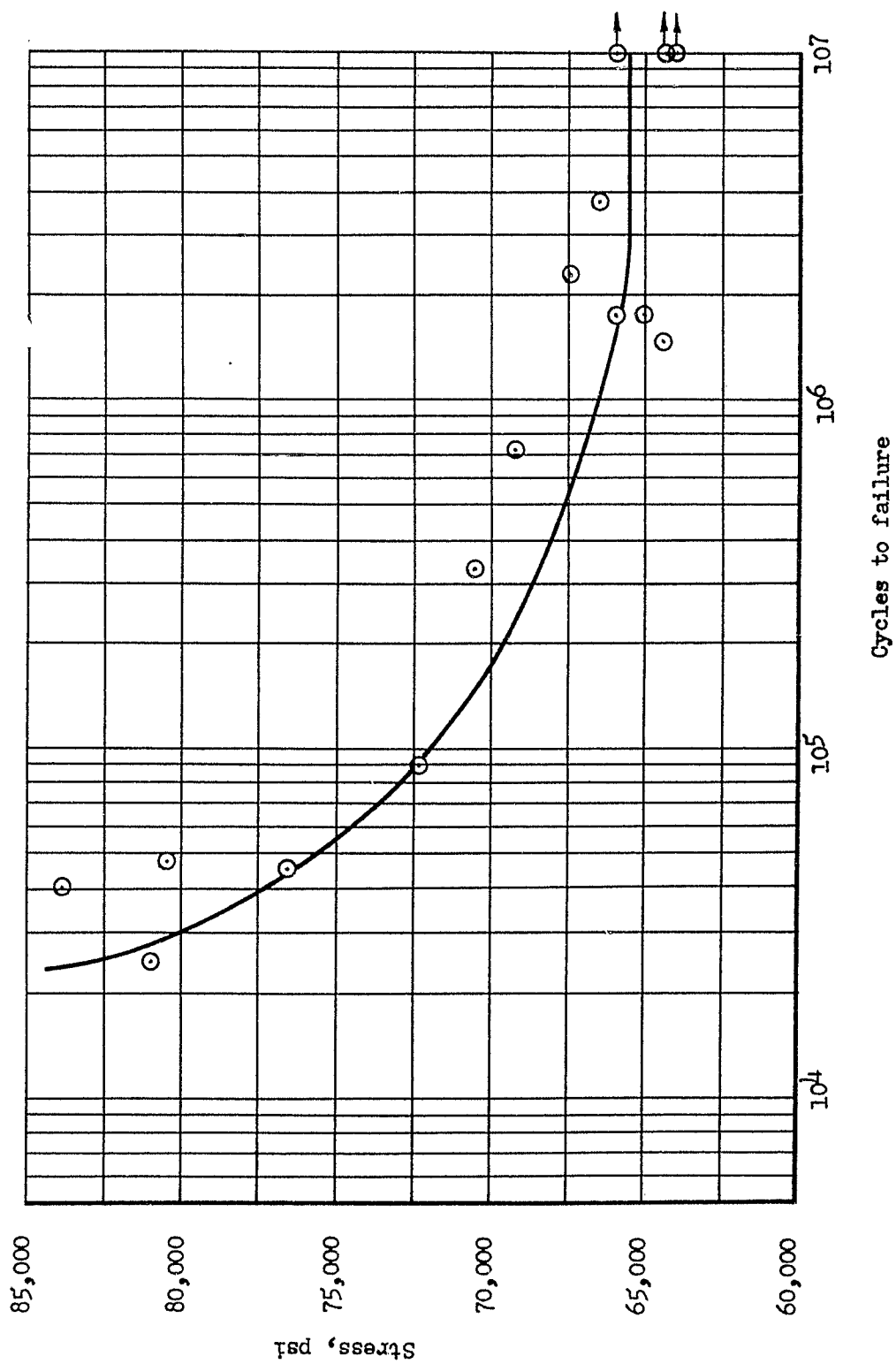


Figure 4.- Dimensions of 1/2-inch-diameter rotating-beam fatigue specimen.



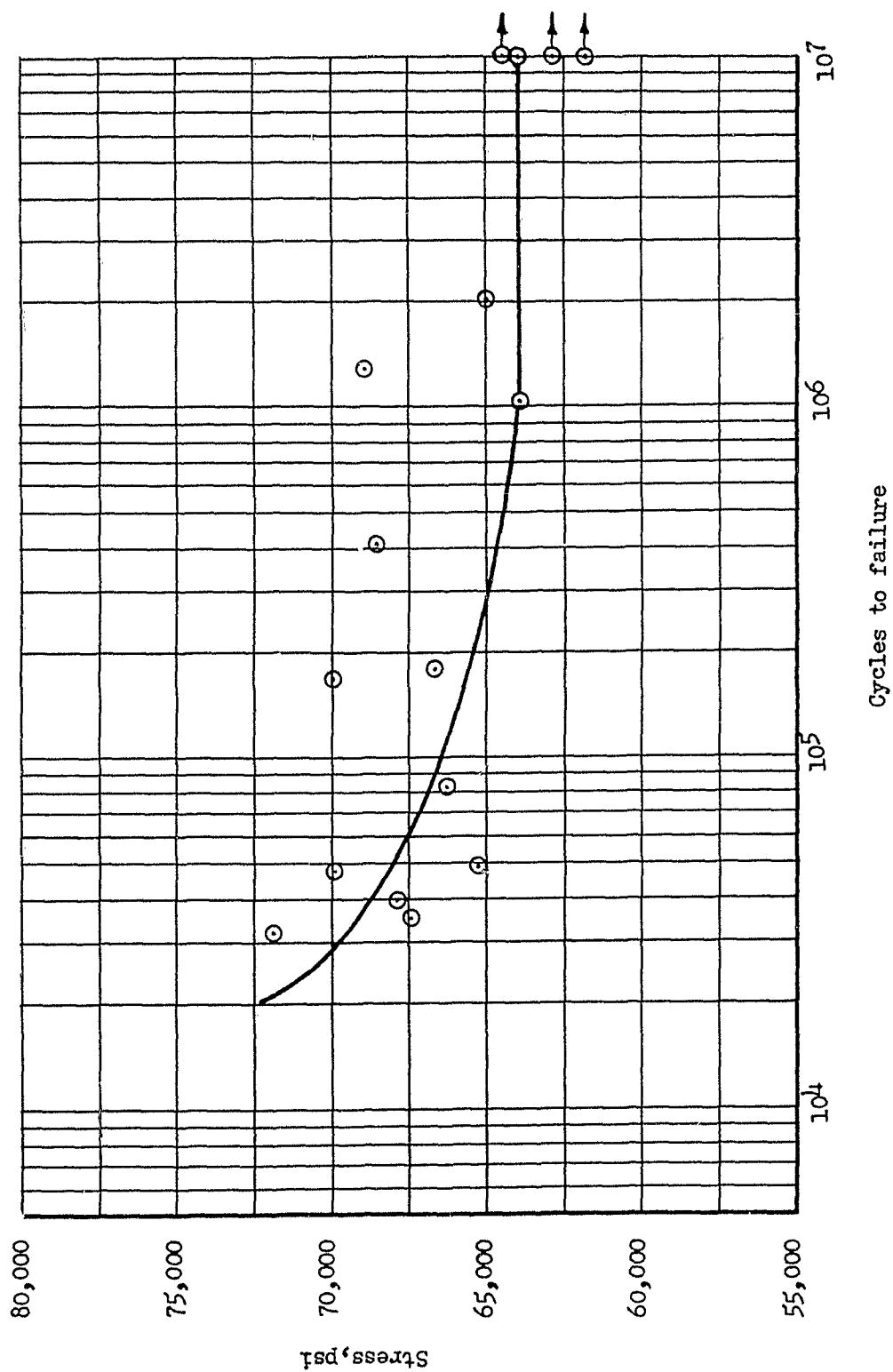
(a) At room temperature.

Figure 5.- Fatigue-test results for 3Mn Complex titanium alloy.



(b) At 200° F.

Figure 5.- Continued.



(c) At 400° F.

Figure 5.- Continued.

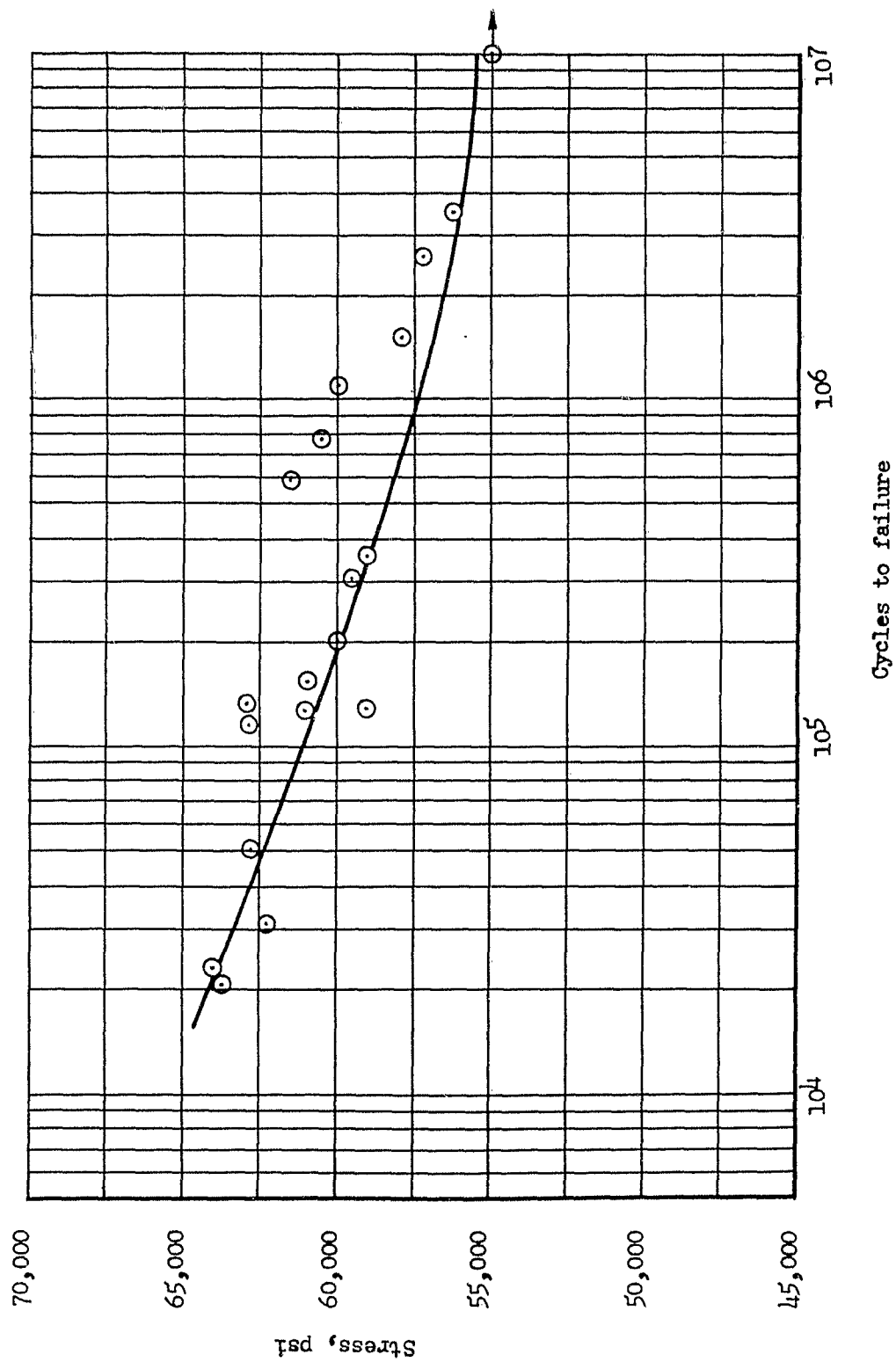
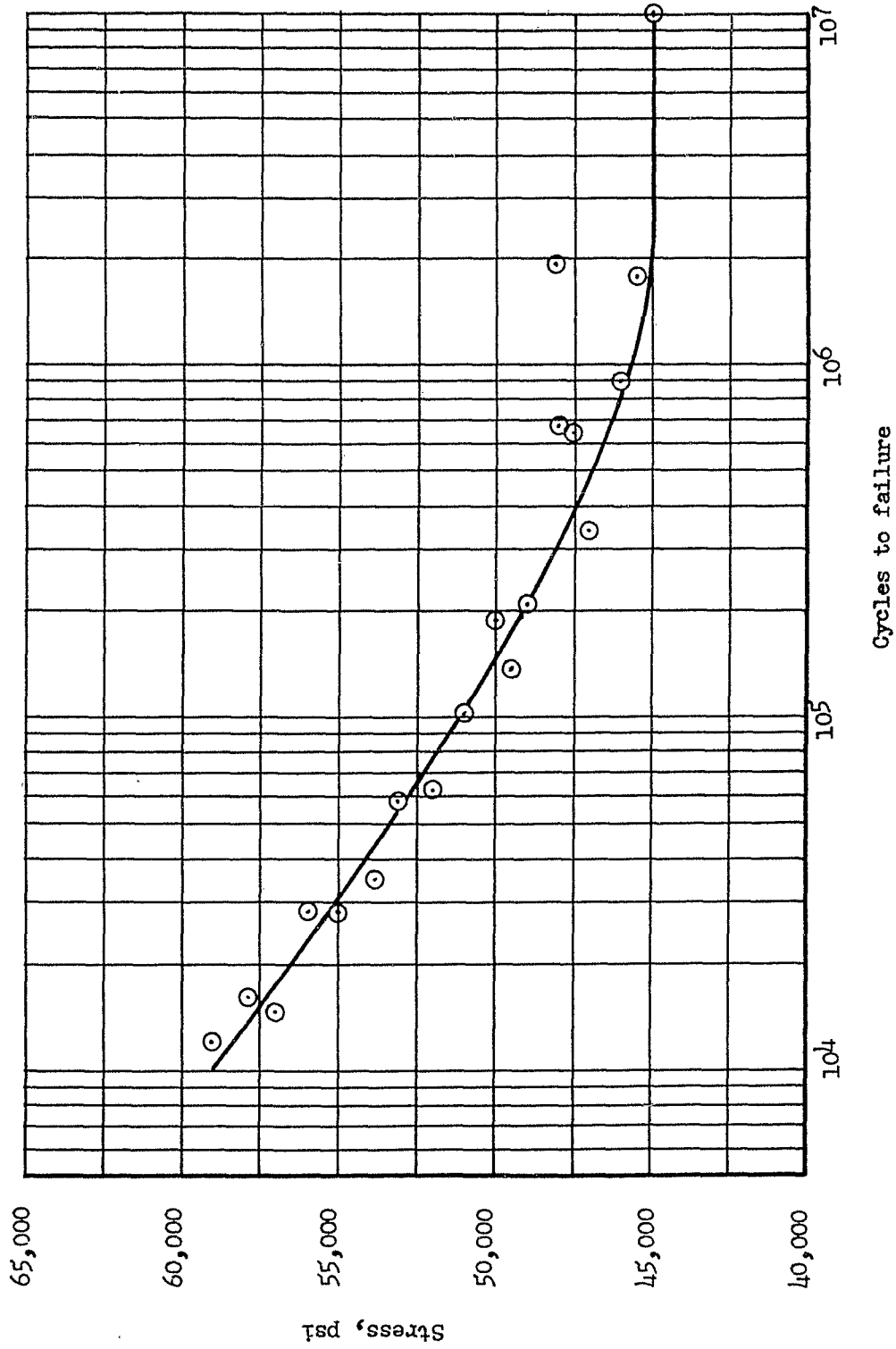
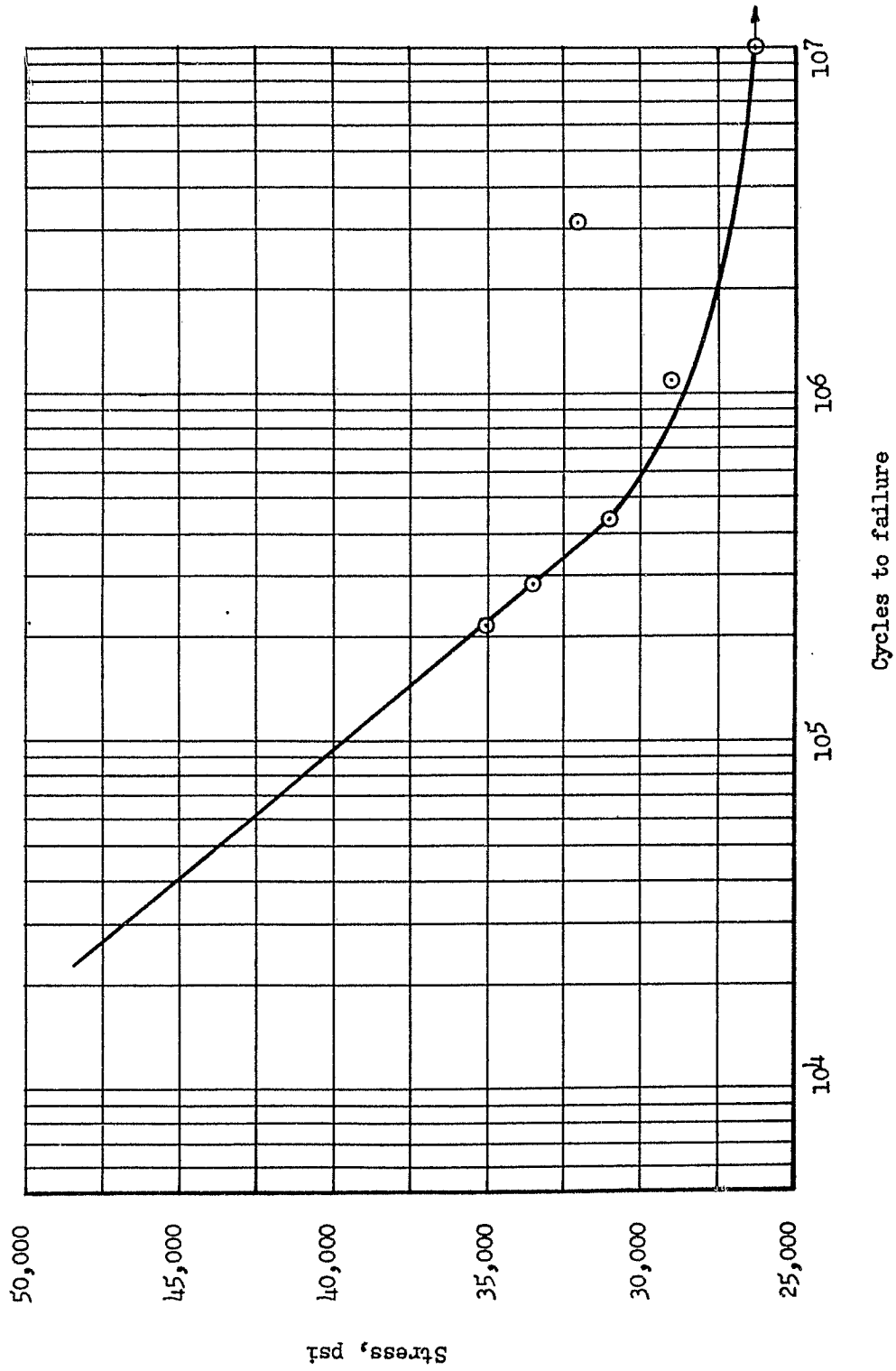


Figure 5.- Continued.



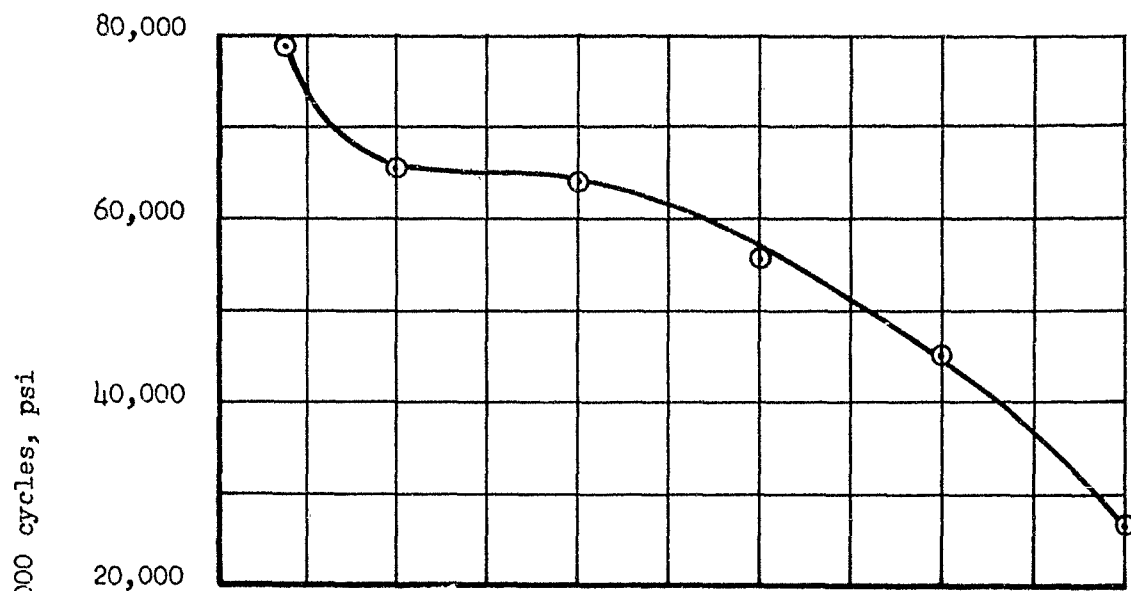
(e) At 800° F.

Figure 5.- Continued.

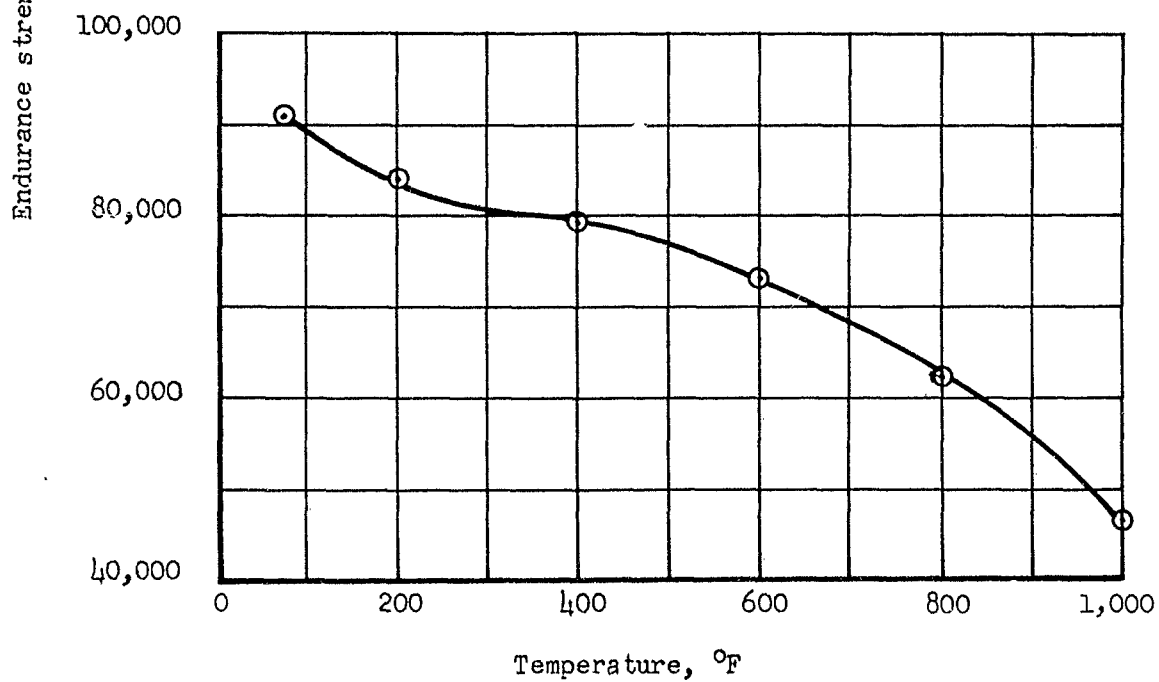


(f) At 1,000° F.

Figure 5.- Concluded.

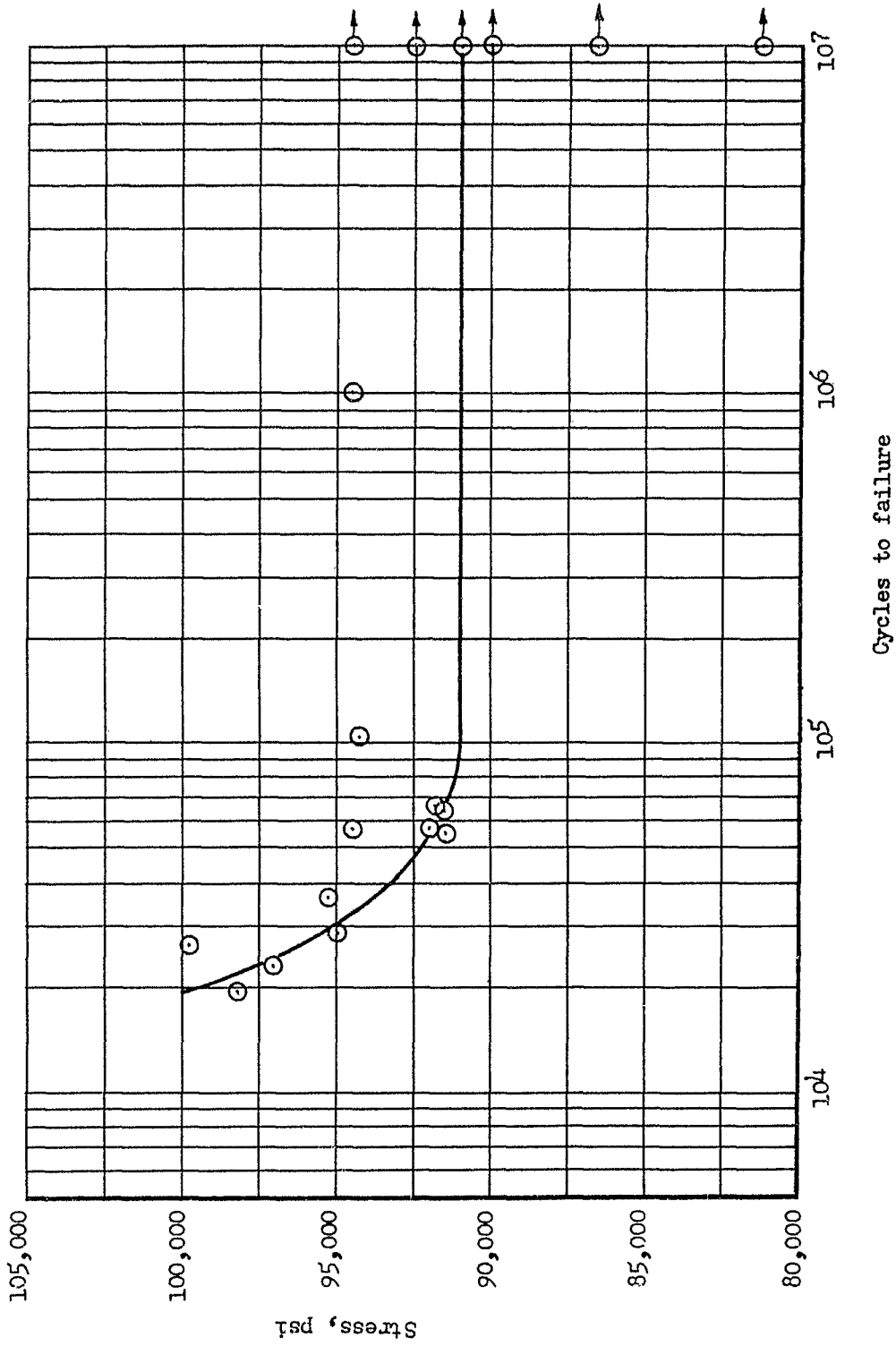


(a) 3Mn Complex alloy.



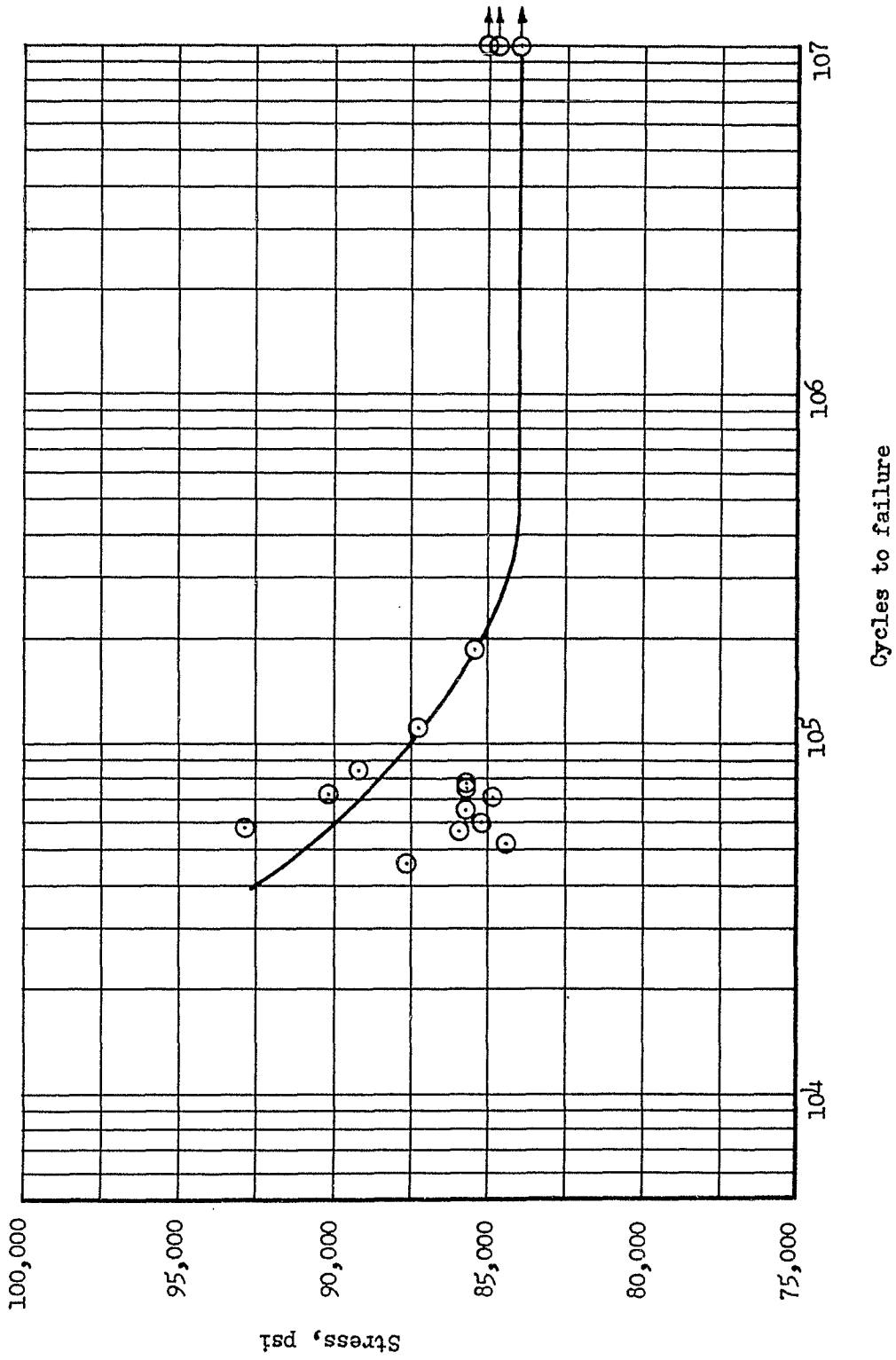
(b) 3Al-5Cr alloy.

Figure 6.- Variation of endurance strength with temperature.



(a) At room temperature.

Figure 7.- Fatigue-test results for 3Al-5Cr titanium alloy.



(b) At 200° F.

Figure 7.- Continued.

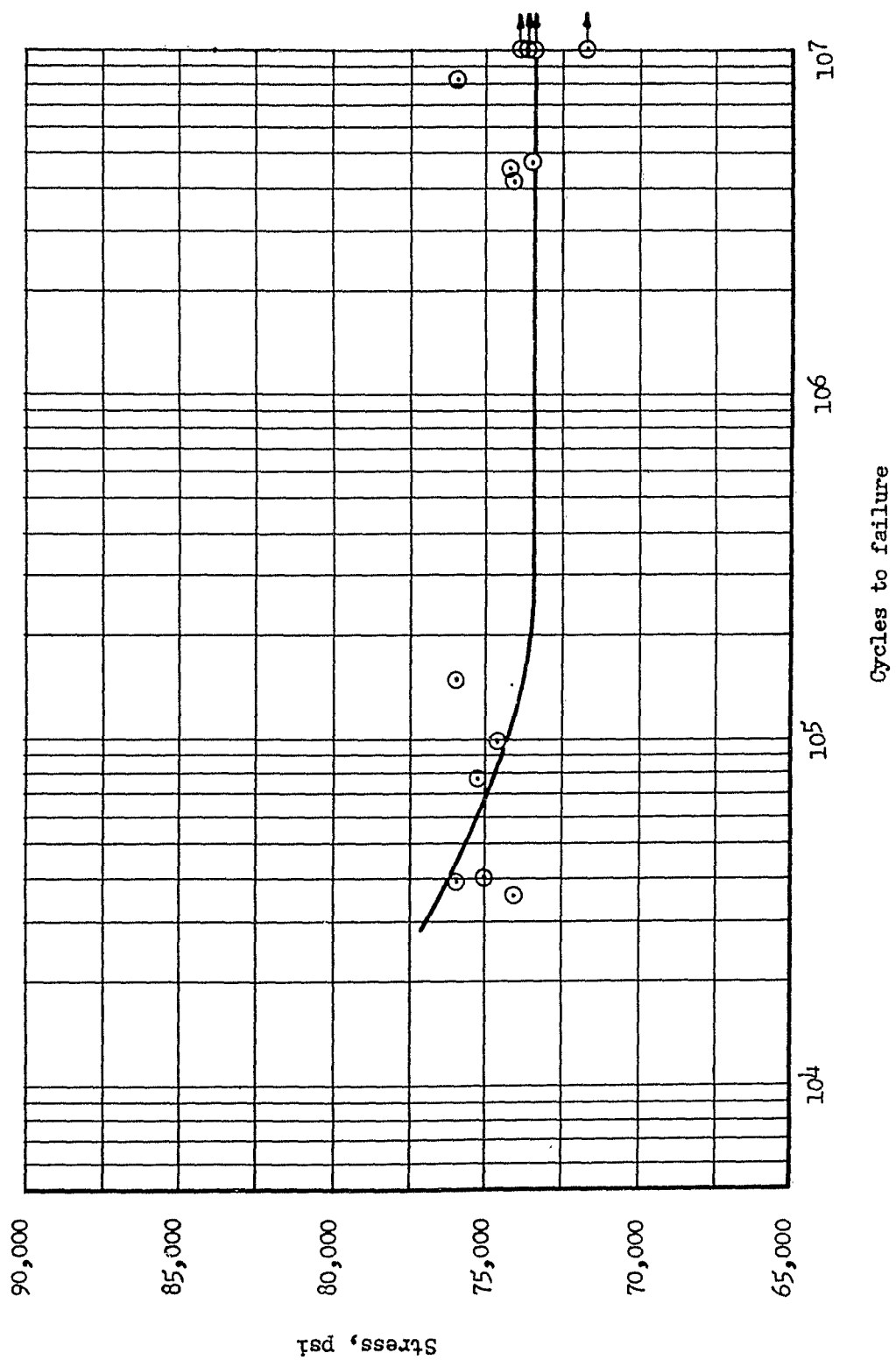
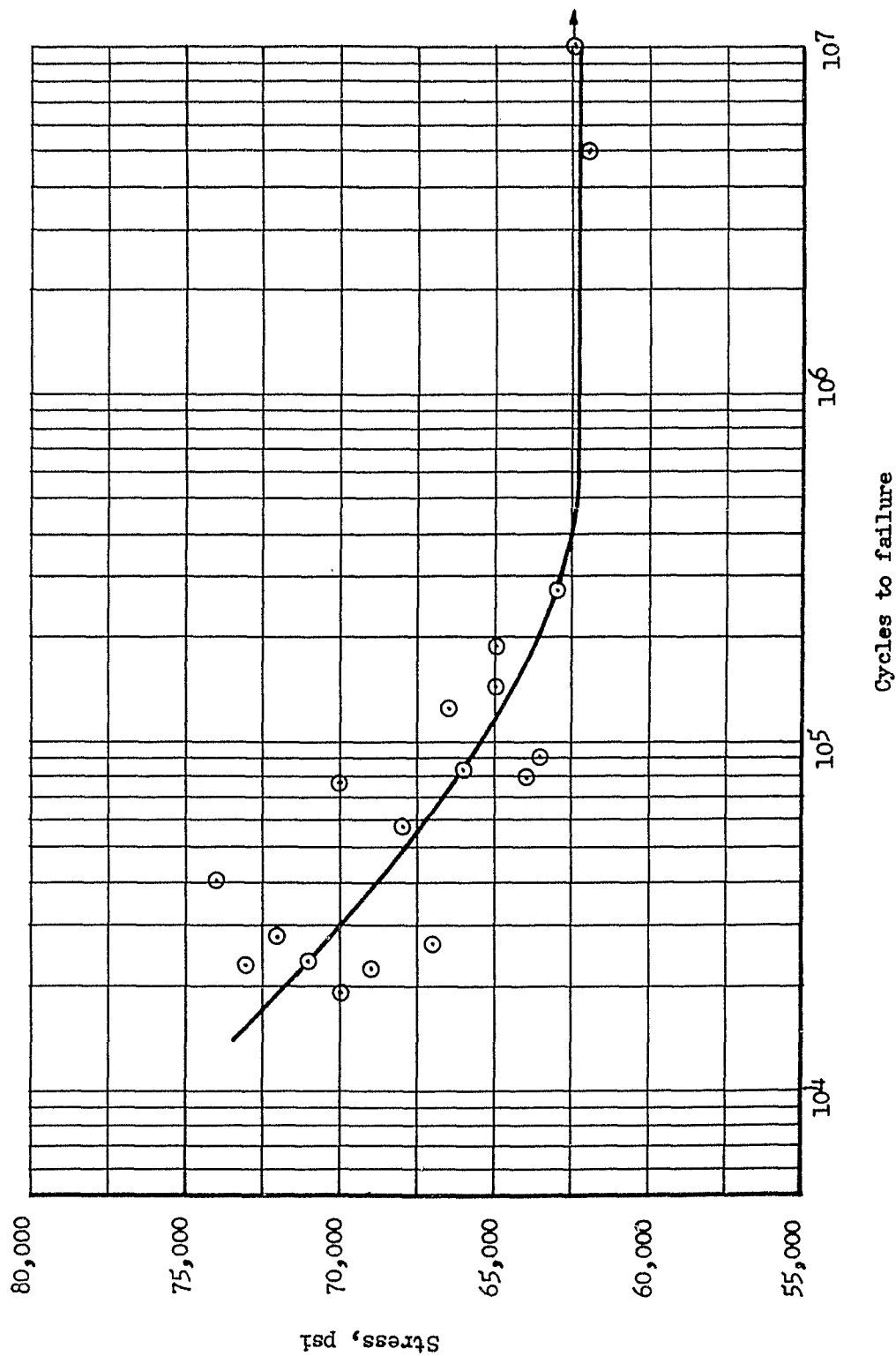
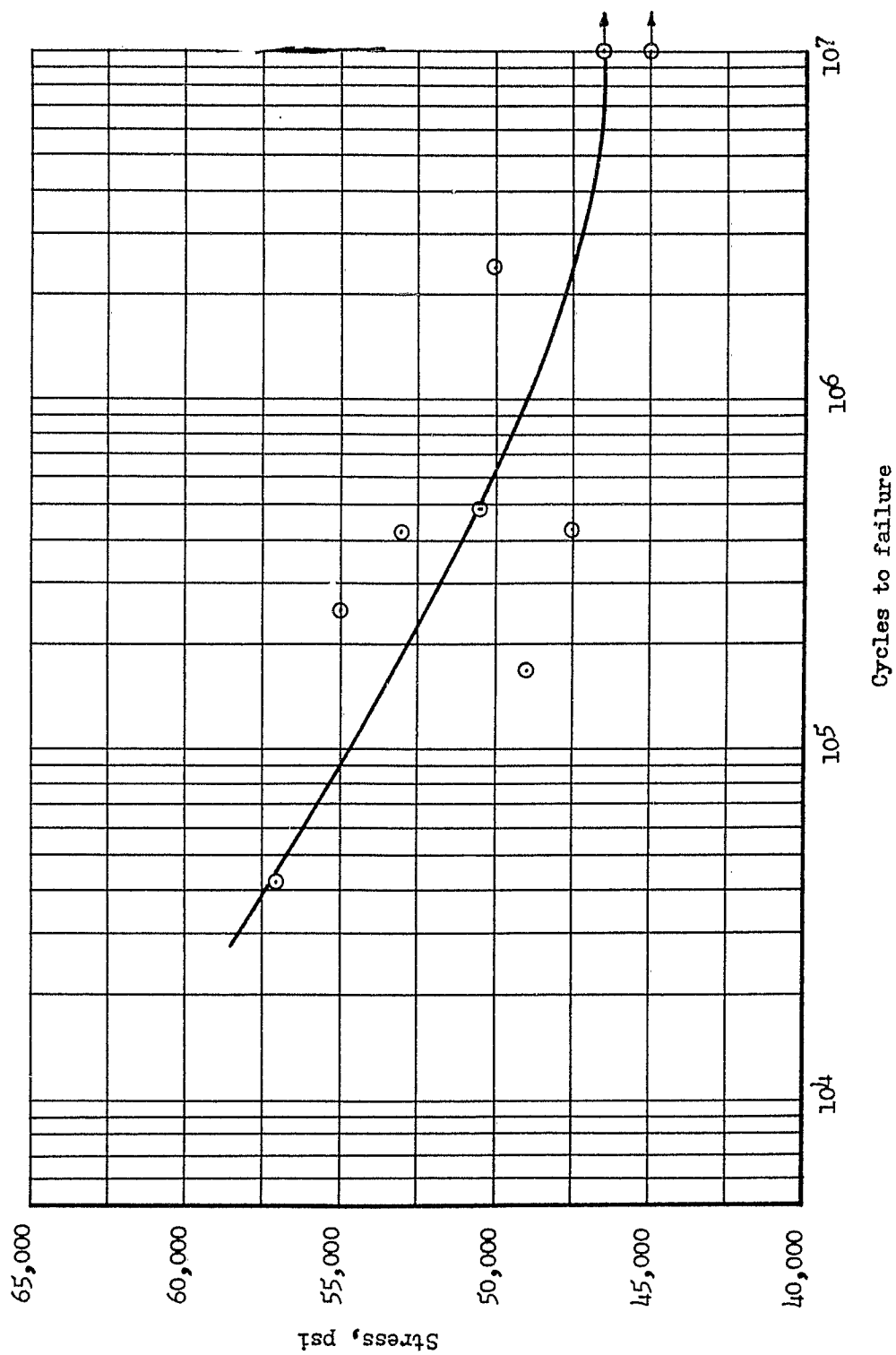


Figure 7.- Continued.





(f) At 1,000° F.

Figure 7.- Concluded.